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2013 Nanotechnology 24 444010

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Nanotechnology 24 (2013) 444010 (7pp)

doi:10.1088/0957-4484/24/44/444010

# Sensor arrays from multicomponent micropatterned nanoparticles and graphene

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Received 17 March 2013, in final form 9 May 2013 Published 10 October 2013 Online at stacks.iop.org/Nano/24/444010

# **Abstract**

A novel approach for multicomponent patterning metal/metal oxide nanoparticles on graphene was developed, which involves region-specific plasma treatment, followed by region-selective substrate-enhanced electroless deposition of Au nanoparticles and solution alkalization of ferrous chloride tetrahydrate in the presence of ammonia into  $Fe_3O_4$  nanoparticles. The resultant  $Fe_3O_4$ /Au multicomponent micropatterned-graphene films were found to be highly selective sensor arrays for detecting low levels of chemical vapor molecules at ppm levels. This novel concept could be applied to the development of various multicomponent patterned nanomaterials for many potential applications, ranging from nanoscale region-specific chemical-/bio-sensor arrays to multifunctional optoelectronic devices.

(Some figures may appear in colour only in the online journal)

# 1. Introduction

Multicomponent micropatterned hybrid nanostructures arranged in a controllable manner have attracted a significant amount of attention due to synergistic properties arising from spatially well-defined different components [1, 2]. Compared to simple single component systems, multicomponent patterns have novel multifunctionalities through the integration of different functional components into designed complex structures [2]. On the other hand, the fabrication of micropatternedgraphene films is essential for the development of graphenebased devices [3]. Owing to its unique properties, such as large surface area, high optical transparency, high electron mobility, and superior electronic properties [4, 5], graphene has been regarded as a nanomaterial with various promising applications, including high-performance nano/microscale integrated sensing devices, new materials for energy conversion and storage, and multifunctional materials for thermal management, to name a few. Many of these and other potential applications require the formation of graphene

films in pre-patterned structures. Among methods reported for producing graphene films, the mechanical cleavage of HOPG [6] and solution exfoliation of graphite crystals [7, 8] are the most widely used methods to produce singleand few-layer graphene sheets of high quality. However, subsequent deposition of materials prepared by these methods often lead to physically adsorbed graphene films of relatively weak interactions with substrates, which may be removed easily during the device fabrication and/or lead to a poor device performance. Therefore, several different methods have been developed for fabricating micropatterned-graphene films, including contact printing using a HOPG stamp made by O<sub>2</sub> reactive ion etching [9, 10], chemical vapor deposition through a mask [11], and electron-beam lithography through a hydrogen silsesquioxane mask [12]. Nevertheless, all of the above-mentioned studies deal with only single component patterning.

The development of graphene micropatterns in which multicomponents are interposed in a controllable fashion is important to multifunctional applications [1, 2]. As far

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as we are aware, however, multicomponent micropatterned-graphene films with different nanocomponents region-specifically interposed in a controllable fashion have not been produced, presumably due to technical difficulties. Here, we report a simple, but effective, spatially controlled method for multicomponent micropatterning graphene films with metal/metal oxide nanoparticles interposed in a controllable fashion. Among many potential applications, the resultant Fe<sub>3</sub>O<sub>4</sub>/Au multicomponent micropatterned-graphene films were demonstrated to be highly selective sensor arrays for detecting low levels of chemical vapor molecules.

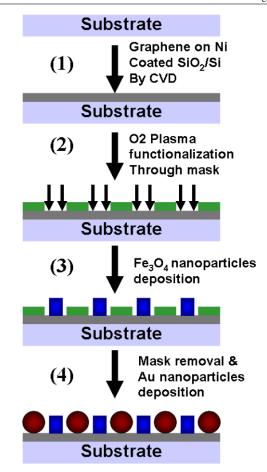
# 2. Experimental section

The fabrication of multicomponent metal-nanoparticle-patterned-graphene films consists of four simple steps, as illustrated in scheme 1.

Chemiresistive materials have been employed to detect toxic industrial chemicals (TICs) and explosives and have been shown to be suitable for gas sensing applications. For example, toxic chemicals (e.g., formaldehyde, CH<sub>2</sub>O) can exist in building infrastructure materials and in the combustion gas of organic materials. Despite the efforts of many research groups in developing gas sensors, challenges still remain related to selectivity, sensitivity, and stability. In order to achieve chemical selectivity and stability for multiple gas sensing applications, in this study, we deposited graphene on a SiO<sub>2</sub>/Si substrate by CVD and region-specifically decorated it with metal and metal oxide nanoparticles for sensing. As shown in scheme 1, a graphene-metal nanoparticle array was fabricated with the oxygen plasma functionalization through a physical mask (e.g., TEM grid), followed by depositing Fe<sub>3</sub>O<sub>4</sub> nanoparticles and the substrate-enhanced electroless deposition (SEED) [11, 12] of metal (e.g., Au) nanoparticles on graphene in the non-plasma-treated areas (note, a very short plasma time has been used to minimize the O<sub>2</sub>-plasma etching effect). Deposition of Fe<sub>3</sub>O<sub>4</sub> nanoparticles was conducted through an alkalization reaction of ferrous ions with ammonium hydroxide (NH<sub>4</sub>OH) in the presence of ammonia to form Fe<sub>3</sub>O<sub>4</sub> nanoparticles within the plasma-treated areas through the hydrophilic-hydrophilic interaction. After removing the mask, the graphene area under the mask can be decorated with Au nanoparticles by using the SEED technique. Metal ions with a redox potential lower than that of graphene can be readily reduced into metal nanoparticles on the graphene layer [11, 12]. The region-selective co-deposition of Fe<sub>3</sub>O<sub>4</sub> and Au nanoparticles into the graphene micropattern can produce region-specifically interposed active sites for selective adsorption and desorption of gas molecules. The newly developed Au and Fe<sub>3</sub>O<sub>4</sub> nanoparticle-decorated graphene hybrid device can be used for selective chemical vapor sensing through monitoring conductivity changes caused by the charge-transfer interaction with gas molecules.

# 2.1. Preparation of graphene film

Chemical vapor deposition was used to develop graphene films by sputtering a thin layer of nickel film (300 nm) on

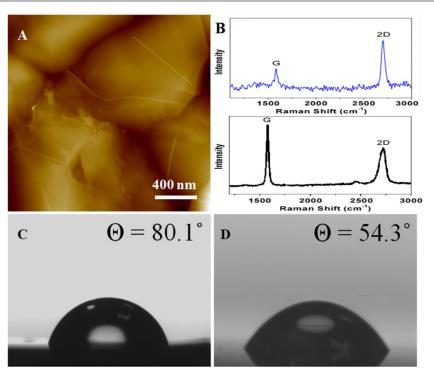


Scheme 1. (A) Fabrication of multicomponent metal-nanoparticle-patterned-graphene films: (1) chemical vapor deposition (CVD) synthesized graphene on 300 nm sputtered Ni catalyst on a SiO<sub>2</sub>/Si substrate, (2) oxygen RF plasma (0.04 mbar, 10 W) was used to functionalize the exposed areas of the graphene with oxygen groups through the TEM grid as a physical mask for 20 s, (3) Fe<sub>3</sub>O<sub>4</sub> nanoparticles are formed on the O<sub>2</sub>-plasma functionalized areas of graphene via the hydrophilic–hydrophilic interaction, followed by the alkalization of ferrous chloride tetrahydrate in the presence of ammonia, and (4) after removal of the TEM grid mask, 5 mM aqueous solution of HAuCl<sub>4</sub> was used for SEED of Au nanoparticles.

a  $SiO_2/Si$  substrate, which was then heated up to  $1000\,^{\circ}$ C within a quartz tube furnace under a high-purity argon atmosphere. Thereafter, a hydrocarbon rich reaction gas mixture (CH<sub>4</sub>:H<sub>2</sub>:Ar, 50:65:200 standard cubic centimeters per minute) was introduced into the quartz tube and kept flowing for 5 min, followed by purging with a flow of Ar for another 5 min. The sample was then rapidly moved out from the furnace under Ar protection.

# 2.2. Oxygen plasma functionalization

Plasma functionalization was carried out on a custom-built plasma apparatus powered at 13.56 MHz, 200 W (Cesar 133 RF power generator) and a pressure of 100 Pa with O<sub>2</sub> flow [13]. The plasma functionalization was applied to the exposed areas of the graphene film through the TEM grid.



**Figure 1.** (A) AFM image of the pristine graphene, (B) typical Raman spectra of the pristine graphene recorded at some spots (*top panel*) and many different locations (*bottom panel*), and contact angle (deg) of H<sub>2</sub>O on graphene (C) before and (D) after the O<sub>2</sub> plasma functionalization.

# 2.3. Multicomponent metal and metal oxide nanoparticles deposition

Nanoparticles of Fe<sub>3</sub>O<sub>4</sub> were synthesized by hydrolysis of an aqueous solution containing iron salts and a base at room temperature in ambient atmosphere [14]. Ferrous chloride (FeCl<sub>2</sub>·4H<sub>2</sub>O, 0.05 M) was used as the iron salt and ammonium hydroxide (NH<sub>4</sub>OH, pH = 12) was used as the base for the reaction [14]. The plasma patterned-graphene film was immersed in the iron salt (ferrous chloride) solution mixed with an aqueous solution of ammonium hydroxide and a precipitate was immediately produced in the plasma-treated areas of graphene through the hydrophilic-hydrophilic interaction. The pH value of the reaction bath was 12. The precipitate was washed with ultrapure water several times to remove excess amine molecules. After removing the mask (scheme 1), SEED was used to deposit Au nanoparticles on the Ni-supported patterned-graphene films by reducing an aqueous solution of HAuCl<sub>4</sub> (5 mM) on the surface of the patterned graphene. The nanoparticle growth process is controlled by diffusion of AuCl<sub>4</sub><sup>-1</sup> from the bulk solution to the non-plasma-treated areas of the graphene/solution interface [15, 16]. Since this is a diffusion controlled process, smaller particles often show a faster growth than larger particles due to relatively more surface-receiving sites for the metal ions [15, 16].

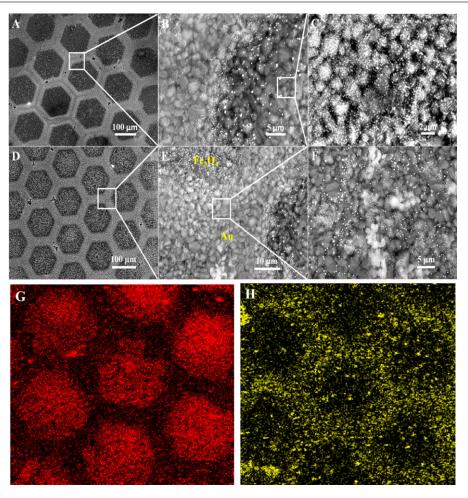
# 2.4. Vapor sensor measurements

The patterned-graphene hybrid sensors were developed by thermally evaporating 2 mm strips of sputter-coated Al about

5 mm apart from each other across the patterned-graphene surface for electrical contacts. The patterned-graphene sensor was then exposed to chemical vapors (e.g., ethanol) in a one-neck flask containing the chemical solvent for resistance measurements with a two-probe dc resistance meter (Dual Display Multimeter, Fluke 45) under ambient conditions [17]. For verification, four-probe dc surface resistance measurements were carried out under the same condition. There was no obvious difference between the two-probe and four-probe results, indicating that good contacts were made with the patterned-graphene film. The resistance of the external wires is insignificant due to their relatively short length [17].

# 3. Results and discussion

A typical atomic force microscope (AFM) image for the pristine graphene is given in figure 1(A), which shows a smooth surface with some wrinkled structures. Figure 1(B) represents the corresponding Raman spectra, showing the co-existence of single-layer graphene (top panel, figure 1(B)) and multilayer (2–5 layers) graphene (bottom panel, figure 1(B)) [18, 19]. The corresponding G-band and 2D-band Raman mapping (not shown) indicates a large area of good homogeneity for the multilayer graphene film, with some randomly scattered single-layer graphene spots. As expected, the O<sub>2</sub>-plasma treatment has made the surface more hydrophilic with respect to the pristine graphene surface (figures 1(C) and (D)).



**Figure 2.** SEM images of the multicomponent patterned graphene. (A) and (D) Hexagonally patterned graphene with  $Fe_3O_4$  nanoparticles within the hexagon and Au nanoparticles deposited in the outer border; enlarged views of the  $Fe_3O_4$  nanoparticle deposited areas (B) and (C) and Au nanoparticle deposited areas (E) and (F) on the multicomponent graphene hexagonal pattern. EDX mapping of (G) Fe and (H) Au.

Figure 2 shows scanning electron microscopic (SEM) images for the Au and  $Fe_3O_4$  nanoparticle multicomponent micropatterns. As can be seen, the resultant micropattern is a close replication of the TEM grid structure with both Au and  $Fe_3O_4$  nanoparticles registered well in respective areas. This is further confirmed by the corresponding EDX mapping of Fe and Au elements shown in figures 2(G) and (H), respectively. Figure 3 shows the atomic force microscope (AFM) images of  $Fe_3O_4$  nanoparticles (A) and (B) in the inner area of hexagonally patterned graphene and Au nanoparticles (C) and (D) in the outer region of the hexagonal graphene frame. Both the SEM and AFM images show aggregated structures formed from smaller particles.

X-ray photoelectron spectroscopic (XPS) analysis was carried out to obtain further information on the surface composition. Figures 4(A) and (B) show the XPS survey spectrum for the graphene/Fe<sub>3</sub>O<sub>4</sub> and multicomponent patterned graphene/Fe<sub>3</sub>O<sub>4</sub>/Au, respectively. The peaks at 710.0 and 723.5 eV in the high-resolution Fe 2p scan can be attributed to the levels of Fe 2p3/2 and Fe 2p1/2 in the multicomponent graphene film, respectively (figure 4(C)). The corresponding high-resolution XPS spectrum for Au

displays a doublet peaks at 87.8 and 83.4 eV for Au 4f5/2 and Au 4f7/2 with, respectively (figure 4(D)). In addition, the sharp peak at 284.5 eV (figures 4(A) and (B)) can be assigned to graphitic carbon while the predominant peak at 541.8 eV is attributable to O1s from the lattice oxygen of  $Fe_3O_4$  (figures 4(A) and (B)).

Decoration of carbon nanomaterials (e.g., CNTs, graphene) with metal or semiconducting particles can lead to sensitive chemical vapor sensors. For instance, Au- and Pt-decorated CNTs elicited a response  $(\Delta R/R_0)$  to NO<sub>2</sub> and NH<sub>3</sub>, which was six to eight times higher than that of pristine CNTs due to a spillover effect at the metal nanoparticles [20, 21]. Electron transport through the newly developed multicomponent metal-nanoparticle-patterned-graphene film can be highly sensitive to adsorbed molecules [20, 21], enabling the development of a sorption-based sensor array for detecting low levels of vapor using low-power electronics. Moreover, the resultant multicomponent graphene film hexagonally patterned with Fe<sub>3</sub>O<sub>4</sub>/Au nanoparticles can be potentially used for region-specific adsorption of analytes from vapor mixtures, and hence region-sensitive sensor arrays. The vapor sensor tests were performed at room temperature using

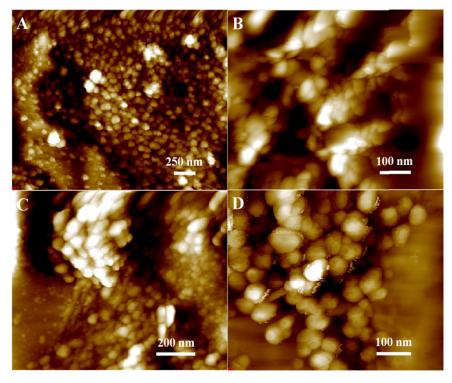


Figure 3. AFM images of  $Fe_3O_4$  nanoparticles (A) and (B) in the inner area of hexagonally patterned graphene and Au nanoparticles (C) and (D) in the outer region of the hexagonal graphene frame.

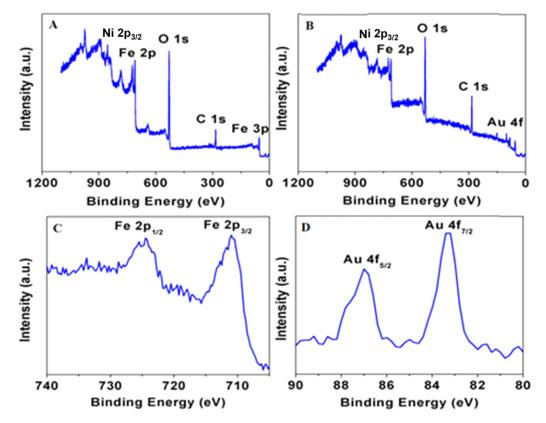
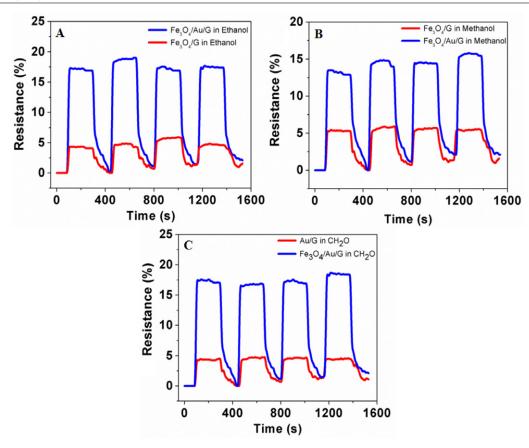


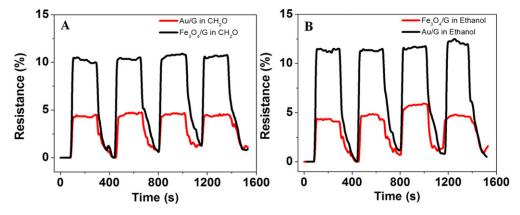
Figure 4. XPS survey spectra of (A) graphene/ $Fe_3O_4$  and (B) multicomponent patterned graphene/ $Fe_3O_4$ /Au, along with the high-resolution spectra of (C)  $Fe_2D_1$ , (D) Au 4f.

ethanol, methanol, and formaldehyde as candidate vapors. Figure 5 shows the response–recovery curves of the  $Fe_3O_4/Au$  patterned-graphene gas sensor array. The sensor showed very

fast response and recovery with a good reproducibility. A comparison study on the sensor performance shows that the patterned multicomponent sensor is at least  $2\times$  more



**Figure 5.** Dynamic response–recovery curves of gas sensors based on the multicomponent Fe<sub>3</sub>O<sub>4</sub>/Au patterned graphene and the corresponding single component sensors to (A) ethanol, (B) methanol, and (C) formaldehyde upon exposure to vapor with time intervals 'on' and 'off'. Resistance (%) =  $(R_{\text{vapor}} - R_0)/R_0 \times 100\%$ , where  $R_0$  and  $R_{\text{vapor}}$  are the resistances of the sensor before and after exposure to ethanol, methanol, and formaldehyde vapor, respectively.



**Figure 6.** Dynamic response–recovery curves of gas sensors based on Fe<sub>3</sub>O<sub>4</sub>/graphene and Au/graphene to (A) formaldehyde and (B) ethanol upon exposure to vapor with time intervals 'on' and 'off', demonstrating the selectivity. Resistance (%) =  $(R_{\text{vapor}} - R_0)/R_0 \times 100\%$ , where  $R_0$  and  $R_{\text{vapor}}$  are the resistances of the sensor before and after exposure to ethanol and formaldehyde vapor, respectively.

sensitive than either of the single component sensors toward vapor molecules, suggesting a synergetic effect arising from the multicomponent micropatterned structure. Furthermore, figure 6 shows the selective sensing of formaldehyde and ethanol by the Fe<sub>3</sub>O<sub>4</sub>/graphene and Au/graphene sensor, respectively. The higher magnitude in the resistance response of the Au-modified and Fe<sub>3</sub>O<sub>4</sub>-modified graphene sensor to ethanol and formaldehyde, respectively, was caused by the specific sensing effect of the Au and Fe<sub>3</sub>O<sub>4</sub> nanoparticles

to the specific chemical vapor. Thus, the multicomponent Fe<sub>3</sub>O<sub>4</sub>/Au patterned-graphene sensor array could be used for sensing both formaldehyde and ethanol even in a region-specific manner. Our preliminary results indicate that similar device performance was obtainable even after transferring the multicomponent patterned graphene without the Ni support onto an insulator silicon wafer, indicating that the Ni film under graphene did not play a significant role, if any, in the sensing process.

## 4. Conclusions

We have, for the first time, demonstrated that regionspecific plasma etching, in conjunction with region-selective metal/metal oxide nanoparticle deposition, can be used to fabricate multicomponent graphene micropatterns with metal/metal oxide nanoparticles interposed within the patterned structure in a controllable fashion, and that the resultant Fe<sub>3</sub>O<sub>4</sub>/Au multicomponent micropatterned graphene films can be used as sensor arrays for detecting low levels of chemical vapor molecules, even in a region-selective manner. The concept developed in this study for multicomponent patterning of nanoparticles on graphene is highly generic, which could be regarded as a general approach toward various multicomponent patterned nanomaterials for many potential applications, ranging from nanoscale (e.g., by e-beam lithography) region-specific chemical-/bio-sensor arrays to multifunctional optoelectronic devices.

# Acknowledgment

This work was partially supported by AFRL/DAGSI (RX2-CWRU-10-1) and DOD-AFOSR-MURI (FA9550-12-1-0037) projects.

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